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# NATIONAL ADVISORY COMMITTEE FOR AERONAUTICS

TECHNICAL NOTE

No. 1770

OFFICE OF NAVAL RESEARCH AND NACA METALLURGICAL INVESTIGATION OF A LARGE FORGED DISC

OF INCONEL X ALLOY

Ву

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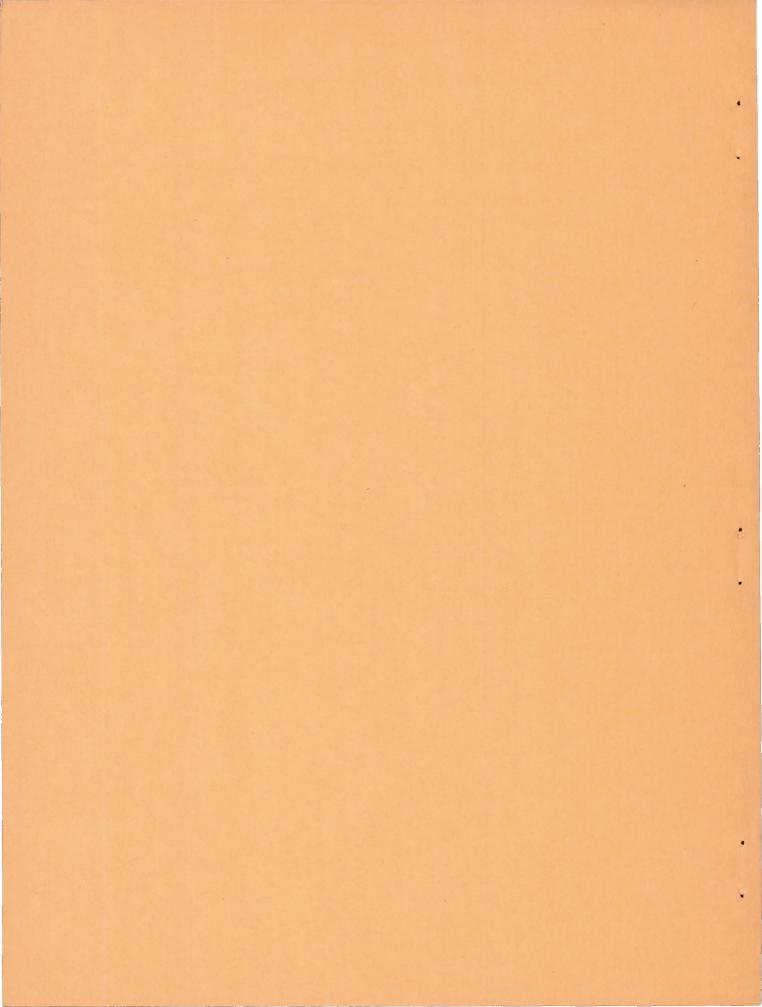
and

J. W. Freeman University of Michigan



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# OFFICE OF NAVAL RESEARCH AND NACA METALLURGICAL

# INVESTIGATION OF A LARGE FORGED DISC

#### OF INCONEL X ALLOY

By Howard C. Cross and J. W. Freeman

#### SUMMARY

The properties of a large forged disc of Inconel X alloy at room temperature, 1200°, 1350°, and 1500° F were studied in order to determine the level of properties obtainable in a forging of the type required for the rotor discs of gas turbines. The disc was tested in the solution—treated and aged condition. The first tests were made on the alloy given a single aging treatment at 1300° F, but it was found that the alloy possessed better properties when given a double aging treatment, first at 1550° F, and then at 1300° F. Most of the test data reported herein were obtained on the alloy given the double aging treatment. The data reported include the results of tensile, impact, rupture, time—deformation, creep, and structural—stability tests.

In general, the Inconel X disc for time periods up to 1000 hours had as high, or higher, properties than other heat-resisting alloys tested at 1200° F. Its superiority decreased with test temperature so that there was relatively little difference between it and other alloys at 1500° F.

It was unusual in that third-stage creep occurred early in all tests, but particularly at 1350° and 1500° F. The disc also retained high ductility at room temperature after prolonged exposure to stress at high temperatures.

#### INTRODUCTION

This report presents the results of a study of the room-temperature, 1200°, 1350°, and 1500° F properties of a large disc of Incomel X alloy tested in the solution-treated and aged condition.

The purpose of this study was to determine the level of properties exhibited by Inconel X alloy in the form of large forgings of the type required for rotor discs of gas turbines. The results obtained previously from investigations on Timken, CSA, EME, 19—9DL, low-carbon N-155, S-590, and S-816 discs of similar size are contained in reports listed as references 1 through 12.

The work on the disc materials is being carried out as part of two correlated programs of research on alloys for gas—turbine applications in progress in this country. The National Advisory Committee for Aeronautics is sponsoring work directed toward the development of improved high—temperature alloys for gas turbines used in aircraft power plants. A concurrent program, formerly sponsored by the National Defense Research Committee, Office of Scientific Research and Development, and now sponsored by the Office of Naval Research, Navy Department, is being directed to the development of alloys for gas—turbine applications in general and, in particular, for both ship and aircraft propulsion. The work herein was performed with the financial assistance of the National Advisory Committee for Aeronautics and the Office of Naval Research, Navy Department.

This report is based on the joint effort of the cooperating research programs. The investigation of these discs for the NACA was conducted at the Engineering Research Institute of the University of Michigan and for the Navy at Battelle Memorial Institute.

#### TEST MATERIALS

The available information describing the disc may be summarized as follows:

Manufacturer:

International Nickel Company, Inc.

Heat number:

Y = 2848 - X

Chemical composition:

C Mm S Si Cr Ni Cb Ti Al Fe Cu
0.05 0.51 0.007 0.39 14.61 73.42 1.04 2.33 0.67 6.90 0.04

Fabrication procedure:

A 4800—pound heat was made in a 6200—pound induction furnace and teemed into an 18— by 18— by 40—inch ingot. This ingot was forged at  $2225^{\circ}$  F to a  $9\frac{1}{2}$ —inch—diameter billet, which was then turned to a  $8\frac{1}{2}$ —inch diameter and cut to a 25—inch length.

This billet was upset by the General Electric Company on a 12,000-pound open frame hammer. Using temperatures between 2225° and 1800° F, the billet was upset in five heats to a disc of 22-inch diameter by  $3\frac{1}{2}$  inches thick.

The solution treatment performed on the as-forged disc by the General Electric Company consisted of heating for 4 hours at 2100° F followed by water-quenching. The disc was sectioned and the various test specimens rough machined prior to aging. At the time that tests on this Inconel X disc were first begun, the recommended aging was a single treatment of 40 hours at 1300° F. Subsequently, low ductility was observed in some of the tests at 1200° and 1350° F on the single-aged material, and additional tests were made on material given a double aging treatment as follows:

1550° F, 24 hours, air-cooled

1300° F, 20 hours, air-cooled

Sampling:

The code number assigned to the disc was NR-99. Figure 1 shows the location of the samples cut from the disc and the code system identifying the coupons. The numerals refer to locations on the flat face of the disc, and the letters refer to the locations through the thickness of the disc.

#### EXPERIMENTAL PROCEDURE

The investigation was designed to provide the following information: (1) The physical properties at room temperature, 1200°, 1350°, and 1500° F which can be expected in large forgings of the Inconel X analysis; (2) the effect of aging treatments on these physical properties; (3) the variation in properties which might be present in various locations in such large forgings; and (4) the change in room—temperature properties resulting from exposure to elevated temperatures under stress for prolonged time periods.

The physical-property data obtained for this large forged disc of Inconel X alloy included short-time tensile properties, impact strengths, rupture test characteristics, and design curves of stress against time for total deformations of 0.1, 0.2, 0.5, and 1.0 percent at 1200°, 1350°, and 1500° F. Creep characteristics were also obtained at 1200°, 1350°, and 1500° F. The data from which the design curves were plotted came from the time-deformation curves of both rupture and creep tests.

The uniformity of the disc was checked by means of a hardness survey, short—time tensile tests, and, to a limited extent, by rupture tests on coupons from representative locations throughout the disc.

The testing procedures used for the short-time tension, stress-rupture, and creep tests were in accordance with the provisions of the A.S.T.M. Recommended Practices E21-43 and E22-41.

#### RESULTS

The data obtained from the Inconel X disc are presented as a series of tables and figures which show the hardness, impact, tensile, rupture, time—deformation, creep, and stability characteristics. The principal results are summarized in figure 2.

# Hardness Survey

Results of the hardness tests on the solution—treated disc after both the first and second steps in the double aging treatment are given in figure 3.

The Brinell hardness as—solution—treated was about 150. The first aging at 1550° F increased the hardness to about 210 and the second aging, to about 285. The surface hardness of the aged disc was slightly higher than for the interior, but the hardness changed little from near the center to the rim.

# Short-Time Tensile Properties

The results of the short-time tensile tests at room temperature,  $1200^{\circ}$ ,  $1350^{\circ}$ , and  $1500^{\circ}$  F are summarized in table I.

Room-temperature tests were made on both 0.505- and 0.250-inch-diameter radial specimens. The 0.250-inch-diameter specimens gave slightly lower tensile strengths than the 0.505-inch bars, but had slightly higher ductility. The results from both types of specimens indicated good uniformity in the disc.

At  $1200^{\circ}$  and  $1350^{\circ}$  F, specimens from the center of the disc gave lower strength values than did radial specimens taken from near the rim. They also tended to have a greater scatter of values. At  $1500^{\circ}$  F, center and rim specimens gave comparable properties.

The average tensile properties of the radial rim specimens as plotted in figure 2 are as follows:

Temperature (°F)	Tensile strength (psi)	0.2-percent- offset yield strength (psi)	Elongation (percent)
75	154,000	97,000	20
1200	117,000	84,500	10
1350	95,500	77,000	7
1500	46,500	43,000	37

# Charpy Impact Resistance

Charpy impact resistance (V-notch) was determined on specimens taken near the rim of the disc. Data are shown in table II and figure 2 for tests at room temperature, 1200°, 1350°, and 1500° F after holding at temperature for a time period sufficiently long to insure a uniform temperature in the specimen.

The Charpy impact resistance increased from about 29 foot-pounds at room temperature to about 60 foot-pounds at  $1500^{\circ}$  F.

# Rupture Test Characteristics

The stress-rupture data for the tests at 1200°, 1350°, and 1500° F are shown in table III, and the rupture strengths obtained from the curves of stress against rupture time in figure 4 are summarized at the bottom of table III.

All the stress-rupture tests were made on 0.250-inch-diameter radial specimens taken from near the rim of the disc. There was no apparent difference in test results from surface and center specimens taken from near the rim.

The rupture strengths at 1200° F for rupture in 100 and 1000 hours were 81,500 and 66,500 psi, respectively. These values are very close to those reported by the International Nickel Company for Inconel X bar stock.

At 1350° F the stresses to produce rupture in 100 and 1000 hours were 53,500 and 37,000 psi, respectively, and at 1500° F, 23,200 and 15,000 psi, respectively. These values are somewhat below those reported by the Nickel Company for bar stock.

Rupture test ductilities were low at 1200° F. The elongation for rupture in 1000 hours was estimated to be about 3 percent. The ductility values at 1350° F were high and held up well as the time to rupture increased. At 1500° F the elongation decreased from 50 percent for rupture in 4.5 hours to 6.9 percent for rupture in 604.2 hours.

# Time-Deformation Characteristics

A convenient method of describing the high-temperature strength of a material is curves of stress against the time required for various total deformations. Such information with the curves of stress against rupture time (and the curves of stress against creep rate) gives design engineers a complete picture of the expected performance of an alloy under conditions of constant tensile stress. The time-deformation data obtained on the Inconel X disc are plotted on semilogarithmic coordinates in figures 5, 6, and 7 for total deformations of 0.1, 0.2, (0.3 at 1200° F) 0.5, and 1.0 percent at 1200°, 1350°, and 1500° F for time periods up to about 2000 hours. Additional curves showing the time of transition from a minimum creep rate to the increasing rate of third-stage creep have been added so as to show when rapid elongation to failure starts.

At 1200° and 1350° F, and to a lesser extent at 1500° F, the curves for the lower total deformations are controlled largely by the proportional limit. In figures 5, 6, and 7 where this is true the curves are dashed to indicate that their location is somewhat variable. It should be emphasized that, in tests at stresses approaching or exceeding the proportional limit of a material, wide differences in initial deformation can be expected because of variations in the proportional limit.

The curves of stress against time for total deformation were plotted from the data in table IV, which also includes the times for total deformations of 2 and 5 percent. The stresses to cause various total deformations in 1, 10, 100, 1000, and 2000 hours, as obtained from figures 5, 6, and 7, are given in table V. These "deformation strengths" are useful numerical ratings of the deformation characteristics and, with the exception of the estimated strengths for 1 hour, are plotted in summary figure 2.

The steepness of the transition—point curves is a danger signal. This material should be watched closely if used at times and stresses beyond the transition—point curve.

# Creep Strengths

Creep rate data, used by many engineers in designs for long periods of service, have been collected from the time-deformation curves of both the stress-rupture and creep tests. Minimum creep rates measured in the rupture tests are included in table III. The detailed creep test data are shown in table VI. The logarithmic curves of stress against creep rate for the rupture and creep tests at 1200°, 1350°, and 1500° F for the Inconel X disc are shown in figure 4.

The creep rates plotted were the minimum rates measured in both the rupture and the creep tests. In the creep tests on this material at 1200° F, in the stress range 45,000 to 50,000 psi, the minimum creep

rates measured were also the final rates at the end of test periods in the range 1500 to 2000 hours. Differing from this behavior, in the creep tests at 1350° and 1500° F, the minimum rates of creep occurred either at the start of the test or during the first 500 hours of the testing period. Following this initial period, increasing rates were observed.

The creep strengths obtained from figure 4 were as follows, and for convenience of comparison the 1000-hour and the extrapolated 10,000-hour rupture strengths are shown here also:

	Michie James	Stress (psi) for indicated properties									
	Temperature 1000—hr rupture strength		0.0001-percent/ hr creep rate	Estimated 10,000—hr rupture strength	0.00001-percent/ hr creep rate						
-	1200 1350 1500	66,500 37,000 15,000	61,000 31,100 15,800	54,000 20,000 9,700	49,000 25,000 13,900						

Incomel X behaves differently from many of the other types of heatresisting alloys included in this program of tests on large discs. The following specific comments should be carefully considered before using creep data for design:

- (1) Incomed X is a strong material and minimum creep rates are low at very high stresses. These minimum creep rates cannot, however, be extrapolated with accuracy on the assumption that a rate of 0.000l or 0.0000l percent per hour is equivalent to 1 percent in 10,000 and 100,000 hours.
- (2) At 1200° F, very little first-stage creep was observed. At stresses of 70,000 psi and higher, however, the tests immediately entered third-stage creep. The time for entrance to third-stage creep increased at lower stresses. It is important to note, however, that third-stage creep was observed in as short a time period as 840 hours at 55,000 psi when the creep rate was only 0.000035 percent per hour. In fact, figure 5 indicates that third-stage creep would occur under the stress corresponding to a minimum rate of 0.00001 percent per hour in only 2000 hours.
- (3) Not only third-stage creep can be anticipated prematurely but also premature rupture. The comparative 10,000-hour rupture strengths given with the previous tabulations of creep strength were well below the creep strengths for 0.000l percent per hour and indicate that fracture would occur at time periods much less than 10,000 hours under stresses corresponding to the creep strengths.

# Stability Characteristics

Some of the completed—test specimens were subjected to tensile, impact, and hardness tests at room temperature after creep testing at 1200°, 1350°, and 1500° F with the results shown in table VII.

After creep testing at 1200° F for 1804 hours, the strength and ductility values measured at room temperature were slightly increased as compared with the values for the material in the original heat—treated condition.

Creep testing for slightly over 2000 hours at 1350° and 1500° F produced a considerable reduction in yield and tensile strengths, with the greatest change at the higher test temperature. Ductility values were somewhat higher after creep testing at 1350° F and practically unchanged after testing at 1500° F. Differing from some other types of alloys, Inconel X consistently maintained a good level of room-temperature tensile ductility and Izod impact strength after long-time creep testing.

Photomicrographs of the Inconel X alloy in the solution—treated and in the solution—treated and aged conditions are shown in figure 8. The very fine precipitate in the aged material is clearly evident.

Figure 9 shows photomicrographs of the structures of creep specimens tested at 1200° and 1350° F. The structures show no significant difference from the as-heat-treated alloy even though tensile properties are somewhat reduced after testing at 1350° F.

The photomicrographs shown in figure 10 for two rupture and creep test specimens tested for 604 and 2160 hours at 1500° F, respectively, definitely indicate a marked change in structure as a result of long—time exposure at this temperature. Appreciable agglomeration of the precipitated phase has occurred and there is a significant difference in structure and degree of agglomeration between the specimens tested for 604 and 2160 hours at 1500° F. This difference in structure accounts for the decrease in room—temperature strength as a result of testing at 1500° F. Because of this lack of structural stability at 1500° F, the use of this alloy at 1500° F should preferably be restricted to service periods over which test data are available and for which the data have shown the deformation at the design loads not to be excessive.

# Comparison of Properties of Inconel X with

# Single and Double Aging Treatments

The first tests on material cut from this Inconel X disc were made on material given a single aging treatment of 40 hours at 1300° F, and this aging treatment was thought to be best. Later, both experimental work by the International Nickel Company and the results of rupture tests at 1200° and 1350° F in the present test program indicated that low

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ductility and low rupture strengths resulted from the use of this single aging treatment. Upon recommendation of the International Nickel Company, a double aging treatment of 24 hours at 1550° F followed by 20 hours at 1300° F was adopted, and most of the tests herein reported were made on the alloy given this double aging treatment. It was thought of interest, however, to indicate some of the results obtained using the single aging treatment for comparison with the data obtained with the double aging treatment.

The data obtained have been summarized and the strengths for Inconel X with both aging treatments are compared in table VIII. It is indicated that Inconel X shows superior rupture strengths and better ductility at 1200° and 1350° F in the double—aged condition. However, at 1500° F the rupture strength of the single—aged material is slightly higher.

Creep strengths for 0.0001 and 0.00001 percent per hour are higher at 1200° and 1350° F for the single-aged material, but the double-aged material shows the better creep strength at 1500° F.

The trend is for superior 100— and 1000—hour deformation strengths at 1200° and 1350° F for double—aged material. At 1500° F, the single—aged material shows the better 100—hour deformation strengths, while for the longer time or 1000—hour strengths up to 1—percent total deformation the double—aged material maintains a slight superiority. These results together with the better ductility indicate that the double aging treatment is the better of the two treatments.

### CONCLUDING REMARKS

The Inconel X disc material developed high tensile and yield strengths when properly solution—treated and aged. Rupture strengths at 1200° and 1350° F were also high. Strengths for total—deformation values in the range of 0.2 to 0.4 percent at 1200° F are largely controlled by minor variations in proportional limit. In tests at all three temperatures of 1200°, 1350°, and 1500° F, the transition to third—stage creep occurred at relatively low deformations and early in the test period. However, Inconel X is such a strong alloy at 1200° and 1350° F that appreciable stresses can be sustained with very low deformations.

Compared with material cut from discs of low-carbon N-155, S-590, and S-816 alloys and tested in various conditions of forging and heat treatment, Inconel X shows generally superior properties at 1200° and 1350° F for time periods up to 1000 hours, which is the extent of the test period for which detailed comparisons can be made. The early transition to third-stage creep for Inconel X suggests extreme caution in extrapolation to longer periods than those for which actual test data are available.

At 1500° F, Incomel X is about equal to or better than any of the other three alloys mentioned for deformations of up to 0.5 percent and time periods up to 1000 hours. Better strengths at 1500° F than those of Incomel X are shown by solution—treated and aged alloy S—816 for a total deformation of 1 percent or more. The observed structural instability of Incomel X at 1500° F and the early transition to third—stage creep are warnings against use of the test data for extrapolation beyond the actual test period.

The high ductility of Inconel X after exposure at high temperature under stress is an unusual and probably desirable characteristic of this material.

Compared with the properties of bar stock of Inconel X given the double aging treatment, the disc material shows slightly lower short—time tensile properties at room temperature and at 1500° F, and slightly lower rupture and creep strengths at 1350° and 1500° F.

Battelle Memorial Institute Columbus, Ohio

and

University of Michigan
Ann Arbor, Mich.,
June 22, 1948

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TABLE I.- SHORT-TIME TENSILE PROPERTIES OF INCONEL X DISC

Specimen	Specimen location	ton CE	strength	Proportional limit		ld strength si)	Elongation in 2 in.	of area	В	elasticity
(a)	(ъ)	( F)	(psi)	(psi)	0.1 percent	0.2 percent	(percent)	(percent)	hardness	(psi)
d <sub>NR</sub> -99-26A	SRR	75	152,800				25.0	24.4	106	
c d <sub>26E</sub>	SRR	75	152,000				27.0	25.9	106	
c d <sub>250</sub>	CRR	75	150,200				27.0	21.3	106	
c d <sub>25D</sub>	CRR	75	151,200				27.0	23.9		
c33X	SRR	75	155,400	75,000	95,400	97,800	19.0	17.0		30 × 10 <sup>6</sup>
°33Z		75	152,600	75,000	91,800	97,400	17.5	15.6		30
°33Y	CRR	75	155,600	72,000	93,000	95,900	21.0	19.2		32
c18A	CRR	75	153,400	78,000	94,400	97,400	23.2	18.4		32
c 39x	SRR	1200	118,500	54,000	79,600	83,600	9.7	14.1		28
°39Z	SRR	1200	119,500	61,000	83,000	86,500	9.2	10.0		25
c39Y	CRR	1200	116,000		81,800	84,900	9.2	11.1		29
c1+OX		1200	114,100	60,000	80,900	84,400	8.0	11.9		29
ellz	SRR	1200	117,750	55,000	79,000		8.5	13.0		26
e34X		1200	113,000		79,500	82,500	12.5	14.8		26
c6x	sc	1200	102,100	60,000	73,800	76,400	11.5	15.2	104	22
c6z	SC	1200	99,000		71,400	73,300	11.7	18.4	104	22
· c7Y	CC	1200	112,800	37,400	69,000	73,100	14.3	15.6	103	25
c8X	CC	1200	93,000		73,050	75,800	8.0	15.2	103	28
chox	SRR	1350	86,200	43,000	71,900	75,400	7.0	7.3		23
°402	1	1350	89,800		73,900	76,600	6.1	10.0		26
elix	SRR	1350	106,000	47,500	76,000	78,300	7.5	11.6		24
e35¥		1350	100,000		75,000	77,500	6.5	10.3		23
c <sub>7X</sub>	sc	1350	76,600	36,000	66,800	68,800	3.0	8.1		
c7Z	sc	1350	81,000		68,500	70,600	4.0	13.0		27
c <sub>412</sub>	SRR	1500	46,700	22,000	38,000	40,200	37.5	38.5		20
°412	1	1500	46,600		39,400	40,500	39.5	44.6		17
c8x	sc	1500	51,200	29,000	44,200	45,900	22.3	26.1		
c8Z		1500	46,400	1	36,000	39,400	43.5	45.5		23

Heat treatment: 2100° F 4 hr water-quenched; 1550° F 24 hr; and 1300° F 20 hr.

DSRR surface-plane radial specimen near rim of disc.

CCR center-plane radial specimen near rim of disc.

SC surface plane near center of disc.

CC center plane near center of disc.

CNavy data.

do.250-in.-diameter specimen; all others 0.505-in.-diameter specimens.

NACA data.

TABLE II. - CHARPY NOTCHED-BAR IMPACT RESISTANCE

AT ROOM TEMPERATURE, 1200°, 1350°,

AND 1500° F FOR INCONEL X DISC

[Navy data: 0.394-in.-square specimens with a 0.079-in.-deep V-notch]

Specimen number (1)	Specimen location	Test temperature (°F)	Charpy impact strength (ft—lb)
NR-99-9YD	Interior	75	29
9YE	Interior	75	28
9YF	Interior	75	32
1ZK	Interior	75	33
1ZC	Surface	75	22
NR-99-1ZG	Interior	1200	38
1ZH	Interior	1200	37
1ZA	Surface	- 1200	37
1ZB	Surface	1200	39
NR-99-1ZI	Interior	1350	45
1ZJ	Interior	1350	47
1ZD	Surface	1350	43
1ZE	Surface	1350	44
NR-99-1ZL	Interior	1500	59
9YA	Interior	1500	54
9YB	Interior	1500	60
9YC	Interior	1500	65
1ZF	Surface	1500	62

Heat treatment: 2100° F 4 hr water-quenched; 1550° F 24 hr; and 1300° F 20 hr.

TABLE III.- RUPTURE TEST DATA AT 1200°, 1350°, AND 1500° F FOR INCONEL X DISC

Specimen number (a)	Specimen location (b)	Test temperature (°F)	Stress (psi)	Rupture time (hr)	Elongation in 1 in. (percent)	Reduction of area (percent)	Minimum creep rate (percent/hr)
°NR-99-27A °27C °28A °27D °28D	SERE CERE SERE CERE CERE	1200	90,000 80,000 70,000 65,000 60,000	36 136 418 1825 2672	3 4 4 4 2	8.7 4.1 3.0 4.0 3.3	0.0085 .0020 .00018 .00007
°NR-99-28C °27E °28B °28E °117-2	CRR SRR CRR SRR SRR	1350	55,000 49,000 45,000 40,000 35,000	86 143 389 751 1177	17 19 16 21	20.5 26.8 18.3 24.0	.0060 .0012 .00028
<sup>d</sup> NR-99-24A d26B d24B d25A	SRR CRR CRR SRR	1500	30,000 25,000 20,000 16,500	4.5 73.6 199.6 604.2	50.0 32.1 18.5 6.9	54.3 34.7 28.1 14.6	e.10 .008 .00045
			Ruptu	re strength			
Tempe	rature			Stress (psi	) for rupture in	1 -	
(0	F)	10 hr		100 hr	1000	) hr	2000 hr
1	1200 1350 1500		· -	81,500 53,500 23,200	37,	500 ,000 ,000	63,000 30,500

<sup>&</sup>lt;sup>a</sup>Heat treatment: 2100° F 4 hr water-quenched; 1550° F 24 hr; and 1300° F 20 hr. <sup>b</sup>CRR center-plane radial specimen near rim of disc.

SRR surface-plane radial specimen near rim of disc.



CNACA data.
dNavy data.

Estimated.

TABLE IV. - DATA FOR STRESS AND TIME FOR TOTAL DEFORMATION AT 1200°, 1350°, AND 1500° F FOR INCONEL X DISC

Specimen number	Temperature	Stress	Initial deformation		Time	(hr) for tot	tal deformat	ions of -				nsition to -stage creep
(a)	-(°F)	(psi)	(percent)		0.2 percent	0.3 percent	0.5 percent	1 percent	2 percent	5 percent	Time (hr)	Deformation (percent)
NR-99-10X	1200	45,000	0.179		c3800							
bloz		47,000	.187		d <sub>1850</sub>							
bllx		50,000	.200									
plox		55,000	.220			1085					860	0.287
p58D		60,000	.242			250	1425	2250			450	.31
p521D		65,000	.267			170	885	1510			230	.31
b28A		70,000	.300				99	265			46	.40
b270		80,000	.420				7	72				
<sup>b</sup> 27A		90,000	•									
enr-99-9x	1350	25,000	.113		1500	1915				/	500	.116
e <sub>12Z</sub>		30,000	.153		505	1570	1000	1240	1490	1720	325	.168
b11Z-2		35,000	.156		126	331	473	635	790		190	.215
p <sup>58E</sup>		40,000	.162		8	80	175	266	356	486	100	.32
b <sub>28B</sub>		45,000	.182				77	128	172	247		
b <sub>27E</sub>		49,000	.192				32	50	68	96		
p580		55,000	.227				10	23	35	50		
*NR-99-9Z	1500	10,000	.050	1515							615	.050
elsx		12,000	.062	775	1700						400	.070
e <sub>25A</sub>		16,500	.104		125		345	440	510	595	170	.22
e <sub>24B</sub>		20,000	.119		12		50	80	86	104	75	.68
e26B		25,000	.168				3	7	16	29	25	3.0
e24A		30,000	.238									

 $<sup>^{\</sup>rm a}$ Heat treatment: 2100° F 4 hr water-quenched; 1550° F 24 hr; and 1300° F 20 hr.

CEstimated by extrapolation of time-deformation curve from 1800 hr. dEstimated by extrapolation of time-deformation curve from 1671-hr. Navy data.

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TABLE V.- TIME-DEFORMATION AND CREEP STRENGTHS AT 1200°, 1350 $^{\circ}$ , and 1500 $^{\circ}$  F FOR INCONEL X DISC $^{a}$ 

Temperature (°F)	Total deformation (percent)		the same of the sa	psi) to c ormation	Creep strength (based on minimum creep rates) (psi)			
( F.)		1 hr	10 hr	100 hr	1000 hr	2000 hr	0.00010 percent/hr	0.00001 percent/hr
p1500	0.2 .3 .5 1.0 Transition	c <sub>50,000</sub> c <sub>70,000</sub> c <sub>87,300</sub>	c <sub>49</sub> ,000 c68,300 78,700	c48,300 63,300 70,000 77,700 67,500	c <sub>47,200</sub> 55,000 61,200 64,500 53,500	<sup>c</sup> 46,000 58,600 60,500	61,000	49,000
b d <sub>1350</sub>	.2 .5 1.0 Transition	c <sub>44</sub> ,000	39,800 55,300 60,000 c <sub>43,100</sub>	35,300 42,700 45,700 40,000	27,000 30,000 31,200		31,100	25,000
<sup>d</sup> 1500	.1 .2 .5 1.0 Transition	c <sub>24</sub> ,500 c <sub>27</sub> ,000 c <sub>29</sub> ,000	c <sub>14,500</sub> 20,600 22,800 24,200	c13,100 16,700 18,700 19,500 18,700	11,100 12,900 14,500 14,800 6,000	11,700	15,800	13,900

<sup>&</sup>lt;sup>a</sup>Heat treatment: 2100° F 4 hr water—quenched; 1550° F 24 hr; and 1300° F 20 hr.

bNACA data.

cEstimated.
dNavy data.

TABLE VI.- CREEP TEST DATA AT 1200°, 1350°, AND 1500° F FOR INCONEL X DISC

Specimen number	Temperature (OF)	Stress (psi)		Initial deformation (percent)	Cree	Creep rate (percent/hr) at -				Total deformation (percent) at-			
(a)	( Fr)		(hr)		500 hr	1000 hr	1500 hr	2000 hr	500 hr	1000 hr	1500 hr	2000 hr	
blox	1200	45,000 47,000 50,000 55,000	1671	0.179 .187 .200 .220	0.000010 .000011 .000017 .000035	0.000008 .000010 .000018 .000055	0.000004 .000010 .000016 .000095		0.182 .188 .213 .273	0.187 .190 .223 .295	0.200 .197 .231 .330		
<sup>c</sup> NR-99-9X	1350	25,000 30,000	2095 d <sub>2193</sub>	.113	.000010 e.00022	.00009	.00014	0.00028	.115 .198	.135 .505	.188	0.295	
c <sub>NR</sub> -99-9Z		10,000	2160 2152	.050 .062	Nil f.000068	.000050 .000068	.000085	.000046 .000135	.048	.070	.099 .160	.130 .235	

aHeat treatment: 2100° F 4 hr water-quenched; 1550° F 24 hr; and 1300° F 20 hr. bNACA data.

CNavy data.

dRuptured, 21.9—percent elongation, 20.9—percent reduction of area.

Minimum creep rate 0.00005 percent/hr up to 325 hr.

Minimum creep rate nil up to 400 hr.

# TABLE VII.— EFFECT OF CREEP TESTING ON ROOM—TEMPERATURE PHYSICAL PROPERTIES OF INCONEL X DISC

Specimen	Prior testin	ng condi	itions		Residual room—temperature properties									
number (1)	Temperature	Stress	Пime	Tensile	(pa	ld strength	limit (pgi)	Elongation in 2 in.	Reduction of area	Izod impact strength	vickers			
	(F)	(psi)	(hr)	strength (psi)		0.2 percent		(percent)		(ft-lb)	hardness			
NR-99-33X 33Z 33Y 18Y	(2) (2) (2)	(2) (2) (2) (2)	(2) (2) (2) (2)	155,400 152,600 155,600 153,400	95,400 91,800 93,000 94,400	97,800 97,400 95,900 97,400	75,000 75,000 72,000 78,000	19.0 17.5 21.0 23.2	17.0 15.6 19.2 18.4	71	306			
NR-99-11Y 10X	1200 1200	50,000 45,000	1537 1804	160,000	102,000	105,000	80,000	14.5	15.8		323			
NR-99-9X	1350	25,000	2 <b>0</b> 95	140,000	75,400	77,800		34.0	30.2					
NR-99-9Z 12X	1500 1500	10,000	2160 2152	125,500	56,500	57,800		22.0	20.2	67	222			

 $^1\mathrm{Heat}$  treatment: 2100° F 4 hr water-quenched; 1550° F 24 hr; and 1300° F 20 hr. 20 riginal condition. Radial specimen near rim of disc.

TABLE VIII. - COMPARISON OF PROPERTIES OF SINGLE- AND DOUBLE-AGED INCONEL X DISC MATERIAL

Test temperature, OF	Room ter	mperature	12	200	13	50	15	500
Condition	Single aged (a)	Double aged (b)	Single aged (a)	Double aged (b)	Single aged (a)	Double aged (b)	Single aged (a)	Double aged (b)
Short—time properties: Tensile strength, psi Elongation, percent Reduction of area, percent	145,000 20 19	154,000 20 18						
Rupture strengths, psi: 100-hr 1000-hr			50,000 to 57,000 35,000 to 46,000	81,500 66,500	46,000 36,000	53,500 39,000	25,500	23,20
Creep strengths, psi: 0.0001 percent/hr 0.00001 percent/hr			46,000 39,000	61,000	c30,000	31,100 25,000	11,200	15,80
100-hr deformation strengths, psi: 0.1-percent deformation 0.2-percent deformation 0.5-percent deformation 1.0-percent deformation Transition			°55,000 °58,000 59,000	°48,300 70,000 77,700 66,800	40,000 43,500 c45,000	35,300 42,700 45,700 40,000	14,200 19,600 21,300 22,200 19,500	c <sub>13,10</sub> 16,70 18,70 19,50 18,70
1000—hr deformation strengths, psi: 0.1—percent deformation 0.2—percent deformation 0.5—percent deformation 1.0—percent deformation Transition			31,000	61,200 61,500 64,500 58,100	37,000	27,000 30,000 31,200	10,000 12,200 13,900 14,500 12,800	11,10 12,90 14,50 14,80 8,00



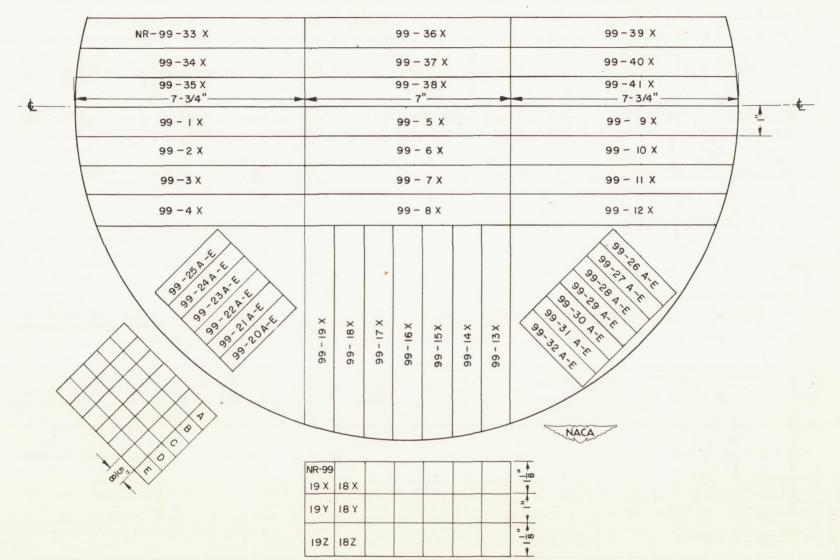


Figure 1.- Location of test coupons in Inconel X disc NR-99.

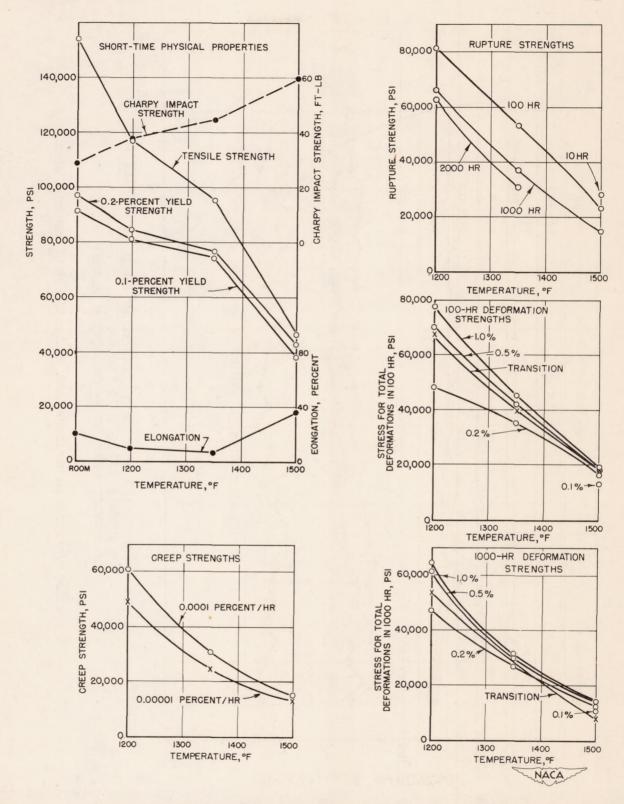


Figure 2.- Summary of properties of the Inconel X disc.

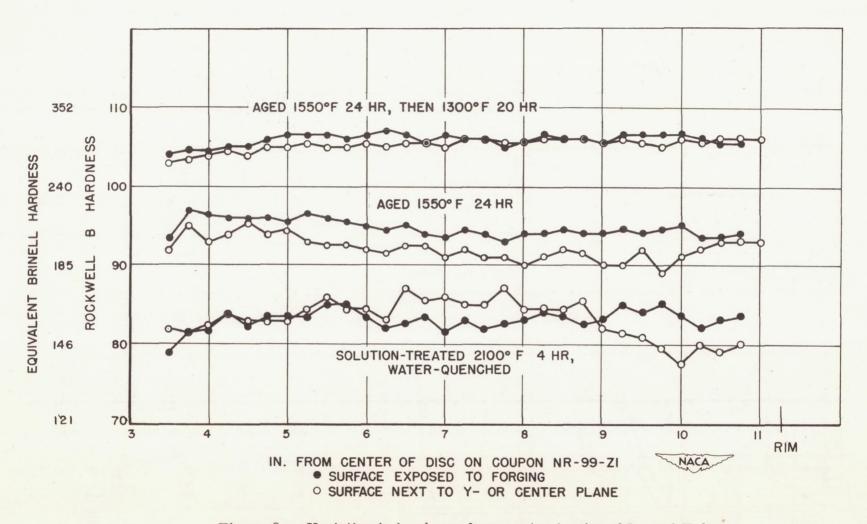


Figure 3.- Variation in hardness from center to rim of Inconel X disc.

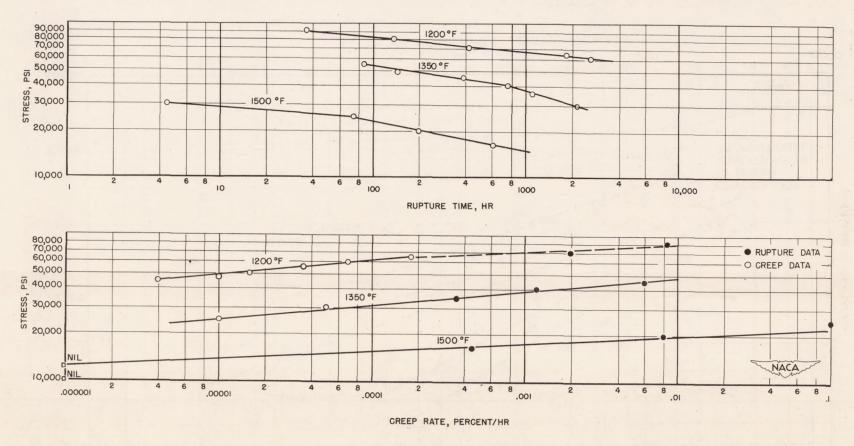


Figure 4.- Curves of stress against rupture time and creep rate at 1200°, 1350°, and 1500° F for Inconel X disc.

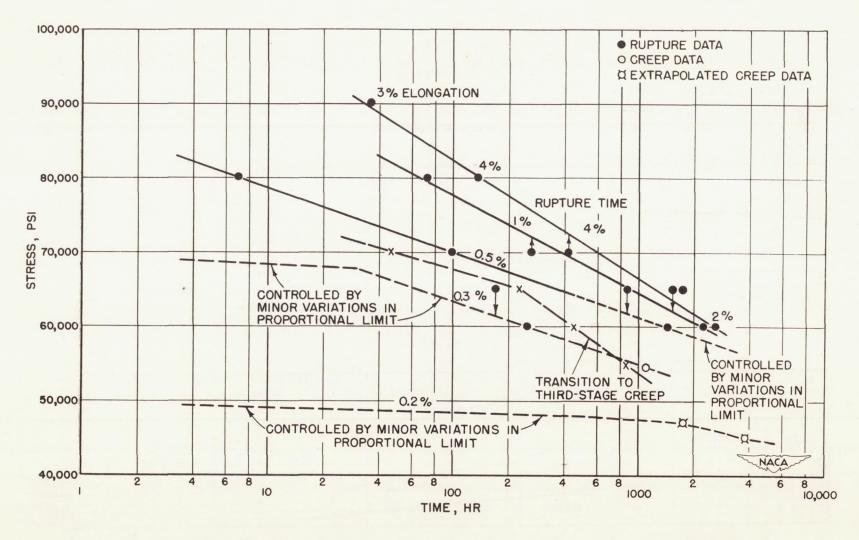


Figure 5.- Curves of stress against time for total deformation at 1200° F for Inconel X disc.

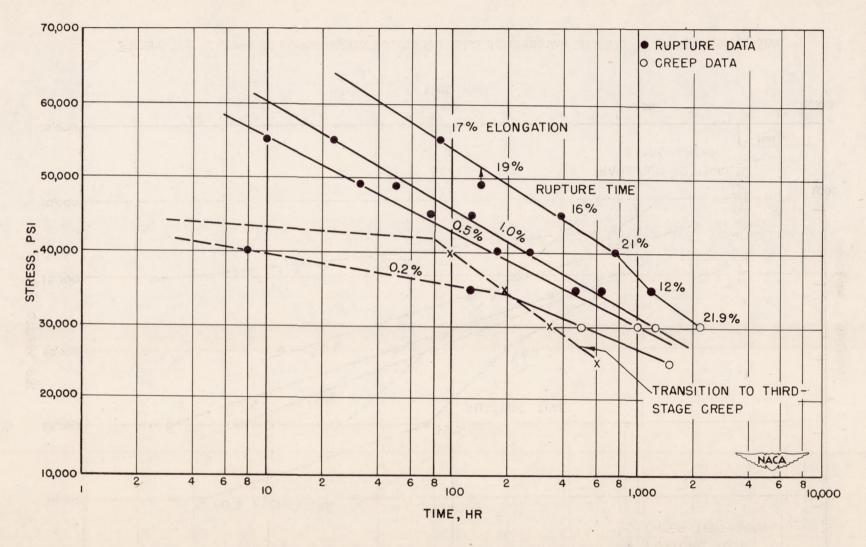


Figure 6.- Curves of stress against time for total deformation at 1350° F for Inconel X disc.

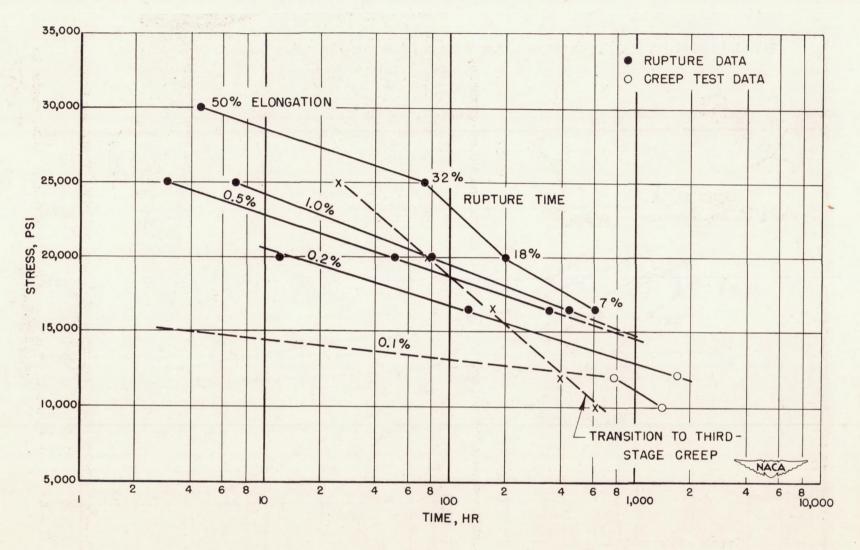
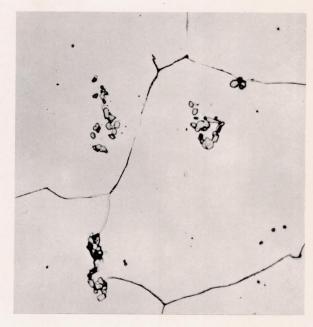


Figure 7.- Curves of stress against time for total deformation at 1500° F for Inconel X disc.

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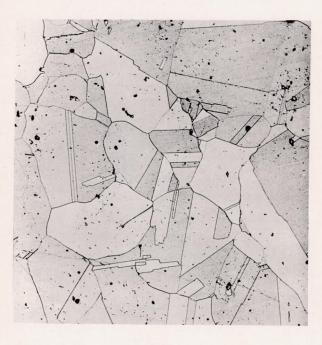




100X

1000X

(a) Solution-treated - heated at 2100° F 4 hours and water-quenched.



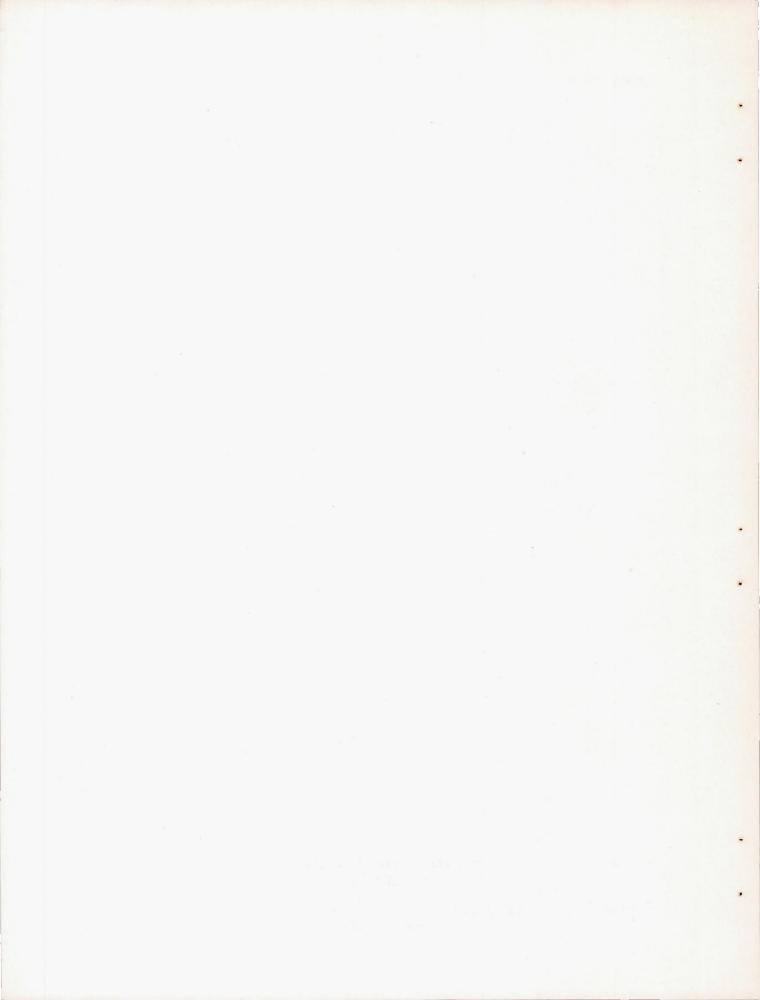


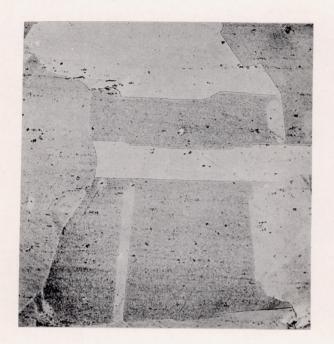
100X

1000X

(b) Solution—treated and then aged 24 hours at  $1550^{\circ}$  F and 20 hours at  $1300^{\circ}$  F.

Figure 8.- Original microstructure of Inconel X disc. Aqua regia in glycerine etch.

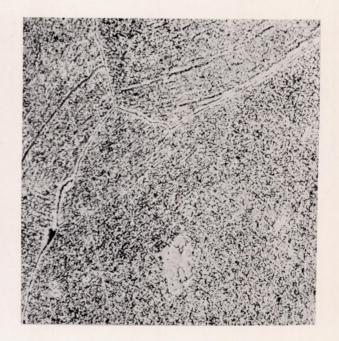






100X 1000X (a) Specimen 11Y; 1537 hours at 1200° F under 50,000 psi.





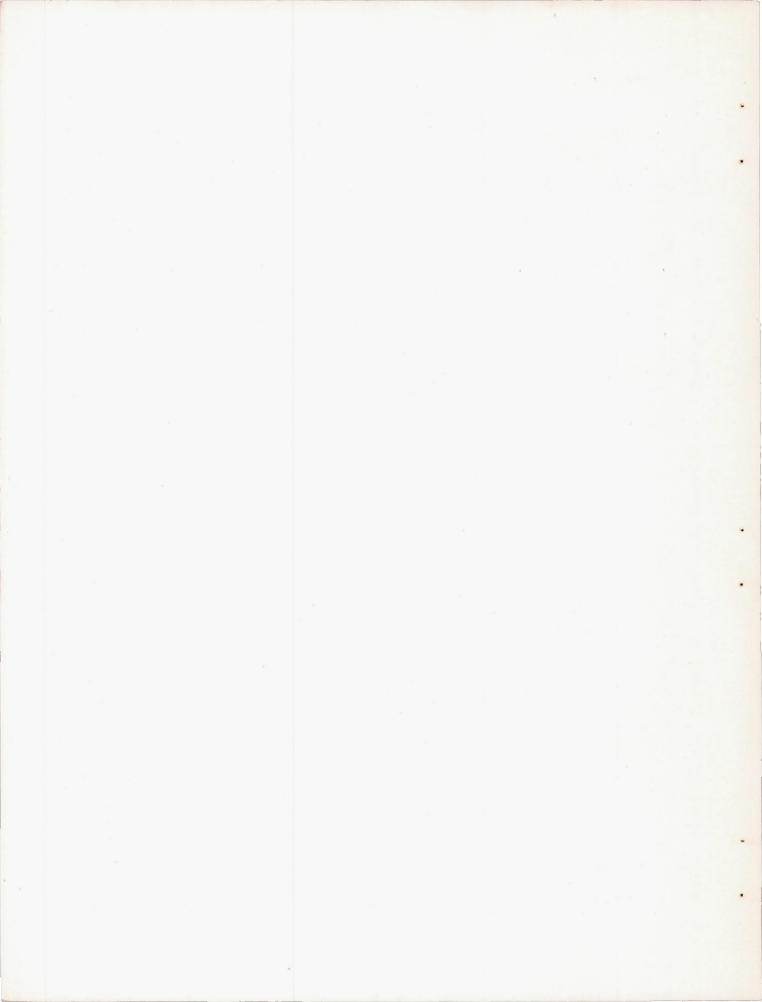
Fracture - 100X

Interior - 1000X

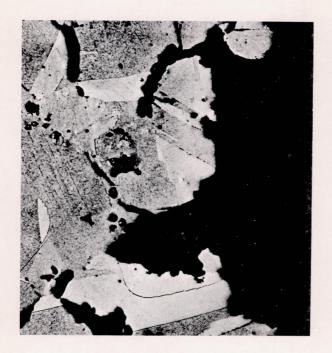
(b) Specimen 28E; 751 hours for rupture at 1350° F under 40,000 psi.

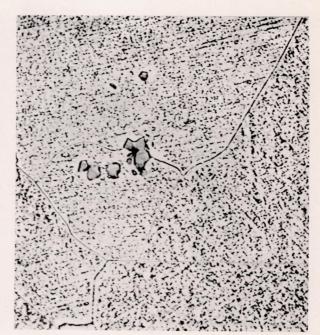
Figure 9.- Microstructures of tested specimens from Inconel X disc.

Aqua regia in glycerine etch.



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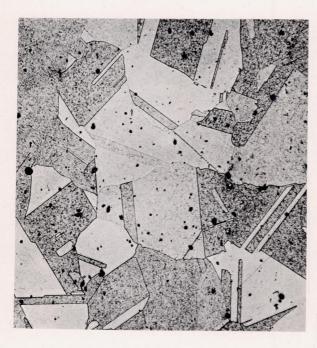


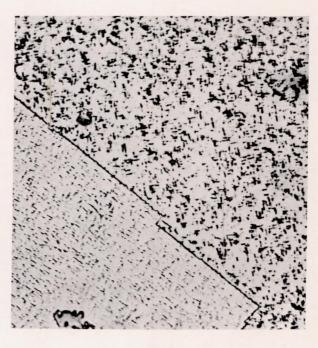


Fracture - 100X

Interior - 1000X

(a) Specimen 25A; 604 hours for rupture at 1500° F under 16,500 psi.





100X

1000X

(b) Specimen 9Z; 2160 hours at 1500° F under 10,000 psi.

Figure 10.- Microstructures of tested specimens from Inconel X disc.

Aqua regia in glycerine etch.